



## LJMU Research Online

**Eslami Mjad, A and Ekere, NN**

**Numerical analysis on thermal crack initiation due to non-homogeneous solder coating on the round strip interconnection of photo-voltaic modules**

<http://researchonline.ljmu.ac.uk/id/eprint/15726/>

### Article

**Citation** (please note it is advisable to refer to the publisher's version if you intend to cite from this work)

**Eslami Mjad, A and Ekere, NN (2019) Numerical analysis on thermal crack initiation due to non-homogeneous solder coating on the round strip interconnection of photo-voltaic modules. Solar Energy, 194. pp. 649-655. ISSN 0038-092X**

LJMU has developed **LJMU Research Online** for users to access the research output of the University more effectively. Copyright © and Moral Rights for the papers on this site are retained by the individual authors and/or other copyright owners. Users may download and/or print one copy of any article(s) in LJMU Research Online to facilitate their private study or for non-commercial research. You may not engage in further distribution of the material or use it for any profit-making activities or any commercial gain.

The version presented here may differ from the published version or from the version of the record. Please see the repository URL above for details on accessing the published version and note that access may require a subscription.

For more information please contact [researchonline@ljmu.ac.uk](mailto:researchonline@ljmu.ac.uk)

<http://researchonline.ljmu.ac.uk/>

# “Numerical Analysis on Thermal Crack Initiation due to Non-Homogeneous Solder Coating on the Round Strip Interconnection of Photo-voltaic Modules”

Alireza Eslami Mjad<sup>a,\*</sup> and Nduka Nnamdi Ekere<sup>a</sup>

<sup>a</sup>School of Engineering, Faculty of Science and Engineering, University of Wolverhampton Shifnal Road, Priorslee, Telford Shropshire TF2 9NN, UK

## Abstract

Solar energy is one of the most widely used renewable energy sources, with photo-voltaic (PV) solar cells/panels now utilized as an important energy source. The strip interconnection between solar cells (used for collecting current from solar cells) is a key PV module component; as poor interconnection reliability can lead to PV module failure. Multi-Busbar is a new type of interconnection which incorporates several round copper wires to help increase the energy conversion and transmission efficiency of PV modules and also to reduce the material costs. The non-homogeneity of the solder coating on the wires (resulting from manufacturing process faults), is one of the main factors that is responsible for the poor connections between the wire and the silver pads; which adversely impacts on the interconnection strength and long term reliability. This paper concerns an investigation of the effect of solder coating non-homogeneity on the thermo-mechanical response of round wires used for PV module interconnections. The study evaluates the two main parameters of non-homogeneity (out of centre value and direction), and also investigates the effect of geometrical parameters. The Extended Finite Element Method in ABAQUS software was used to determine the micro-crack initiation temperature and location for a given joint design. The results show that the cracking temperature is most affected by the direction of solder coating non-homogeneity and the downward vertical direction of out of centre positioning of copper in the solder coating leads to the most reduction in cracking temperature (up to 21%).

**Keyword: PV Module, Multi-Busbar interconnection, Non-Homogeneity of Solder Coating, Crack Initiation, Reliability.**

## 1. Introduction

Ribbon interconnections are one of the main components of PV Modules which connect the individual crystalline silicon solar cells and transmit and conduct their electrical current to external circuits. The design of the ribbon interconnection therefore has a direct influence on the efficiency of

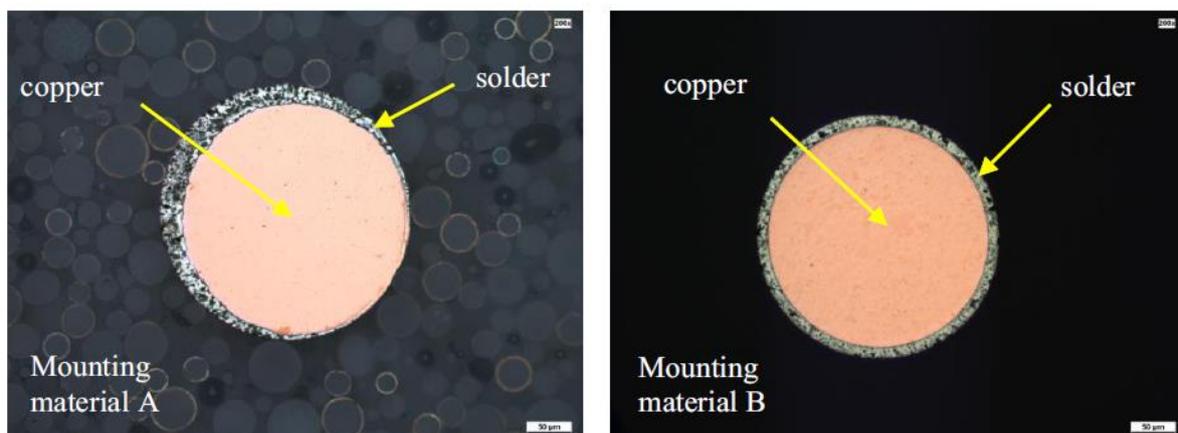
---

\* Corresponding author. Tel.: +44 (0)1902 321 703; fax: +44 (0)1902 321 459.

E-mail addresses: a.e.majd@wlv.ac.uk (Alireza Eslami Mjad), ndyekere@wlv.ac.u (Ndy Ekere)

34 the PV module (in terms of energy conversion and transmission) and the long term reliability. The  
35 configuration of the interconnection influences the Shading on the PV Cells and the partial shading  
36 loss (with partial shading, one part of the solar panel generates lower amount of energy as compared  
37 to the other non-shaded part) reduces the performance and efficiency of PV Modules (Pareek et al.,  
38 2017). Many efforts focussed on developing new joining materials have been reported, such as that on  
39 electrically conductive adhesive (ECA) joints or conductive paste (CP)-assisted low-temperature  
40 soldering (CALS) as an adhesive layer to connect the strips to the cells (Song, et al., 2019) and  
41 although this leads to a cheaper interconnection solution (use of less solder materials), the  
42 interconnections do not have the required high shear strength.

43 The Multi-Busbar (MBB) Connector is one of the innovative concepts for strip interconnection  
44 and solar cell tabbing. The aim of the MBB design is to increase the number of busbars and to reduce  
45 the busbars' width and finger cross sections (Schneider et al., 2006). In comparison with conventional  
46 interconnection method, the use of the MBB technology leads to a reduction in the series resistance  
47 losses, due to the increased number of current paths and the associated uniform current distribution  
48 (Schindler et al., 2013) and (Walter et al., 2014). For example, by using 15 wires instead of three  
49 traditional rectangular busbars, the finger length can be reduced from 25mm to 5mm and the width of  
50 the finger can also be reduced from 50  $\mu\text{m}$  to 17  $\mu\text{m}$  (Braun et al., 2013). This reduction in Ag-paste  
51 area results in bringing down the cost of PV modules, by reducing the consumption of silver by up to  
52 89%. In addition, the use of rounded edge wires in the MBB design helps to reduce the shadowing  
53 loss as most of the incident light can be reflected onto the wafer surface; helping to increase the power  
54 output as well as the efficiency of PV (Braun et al., 2012). However, the MBB Connector has one  
55 unresolved design/manufacture challenge; the problem associated with the non-homogeneity of solder  
56 coating that is produced around the Cu-wire which can result in weaker solder joints and  
57 interconnections (Walter et al., 2014). Figure 1, shows the cross-sectional views of round copper  
58 wires with homogeneous and non-homogeneous solder coating.



59  
60 Figure 1 Round wire with non-homogeneous (left) and homogeneous (right) solder coating (Walter et al., 2014)

61 Thermal stress due to coefficient of thermal expansion (CTE) mismatch between interconnection  
62 materials and PV module materials can lead to early interconnection failure and adversely impact on  
63 the reliability of the PV module interconnection (O. Ogbomo et al., 2018) and (Zarmai et al., 2016).  
64 Indeed, in high temperatures, the weaker MBB solder joint connection area subjected to the thermal  
65 stress can develop micro cracks. Previous studies show that the cracks are developed mainly in the  
66 contact surface of the strip interconnection and the contact condition can adversely affect the  
67 performance of the whole module in terms of power output (Jeong, et al., 2012) and (Itoh et al.,  
68 2014). Indeed, the crack developed induces contact resistivity between the Cu ribbon interconnection  
69 and cell; resulting in cell-to-module (CTM) loss, hot spot and eventually the disconnection of the bus  
70 bar line which consequently results in DC arc (Itoh et al., 2014) and (Tae-hee Jung, 2014).

71 Rendler et al. have investigated the deformation of cell and thermomechanical stress in both cell  
72 and interconnecting wire of MBB PV modules. They found by using lower diameter of wires and by  
73 reducing the Young's modulus or the yield strength of the copper, the thermomechanical stress in a  
74 solar cell is decreased. They also recognized that the maximum stress in the wires occurs at the edge  
75 of the outermost contact pads on both sides of the solar cell (Rendler et al., 2016, 2018). This  
76 important region of the solder joint material is where the Intermetallic compound (IMC) is formed.  
77 Intermetallic compounds (alloys) form whenever two different metals are soldered together and grow  
78 as solid phases during the solidifying of solders on the interface between the solder alloy and its  
79 bonding pads (Pecht, 1993). Generally, the thermal fracture of a solder connection due to crack  
80 propagation is divided into two distinct modes. Firstly, inside a solder joint (solder-controlled  
81 fracture) due to the growing grain size of solder and decreasing bonding strength during thermal  
82 cycles, a crack progresses at the interface of large grain. Secondly, at the interface of the solder with  
83 interconnection material layers by generating Intermetallic Compounds (IMC) layer such as  $\text{Cu}_5\text{Sn}_6$   
84 and  $\text{Ag}_3\text{Sn}$  due to the dissolution of Ag/Cu in the solder and the formation of a brittle layer (Itoh et  
85 al., 2014). Experimental observations show that the fractures and straight crack path are located on the  
86 component side of the solder interconnections (beneath die edge) (Li et al., 2012). Consequently, it is  
87 obvious that the IMCs layers in the solder region boundaries play a significant role in the nature of  
88 failure and subsequently life time and reliability of solder interconnection. The focus of this study is  
89 on the effect of non-homogenous coatings on the strength of brittle micro-cracking in the Intermetallic  
90 Compound (IMC) of solder joint in PV module's interconnection.

## 91 **2. Methodology and Simulation**

92 In this study, we investigate the effect of out of centre of solder coat around copper wire on the  
93 crack initiation temperature. Extended Finite Element Method (XFEM) in ABAQUS 6.17 has been  
94 utilized to analyse the hot temperature crack initiation (ABAQUS Theory Manual, 2017). XFEM is  
95 based on the partition of unity method, and this numerical technique extends conventional finite

96 element method and allows local enrichment discontinuous functions to be combined with the  
97 conventional finite element approximation (Sivakumar & Maji, 2016). One of the main advantages of  
98 this method is avoiding any need for re-meshing or geometric crack modelling in numerical  
99 simulation, while generating discontinuous fields along a crack and around its tip (Mohammadi,  
100 2008). Unlike conventional FE methods, with the XFEM method there is no need for specifying the  
101 location of the crack beforehand. This means that XFEM models a crack within an element as an  
102 enriched feature by adding degrees of freedom in elements with special displacement functions  
103 (ABAQUS Theory Manual, 2017).

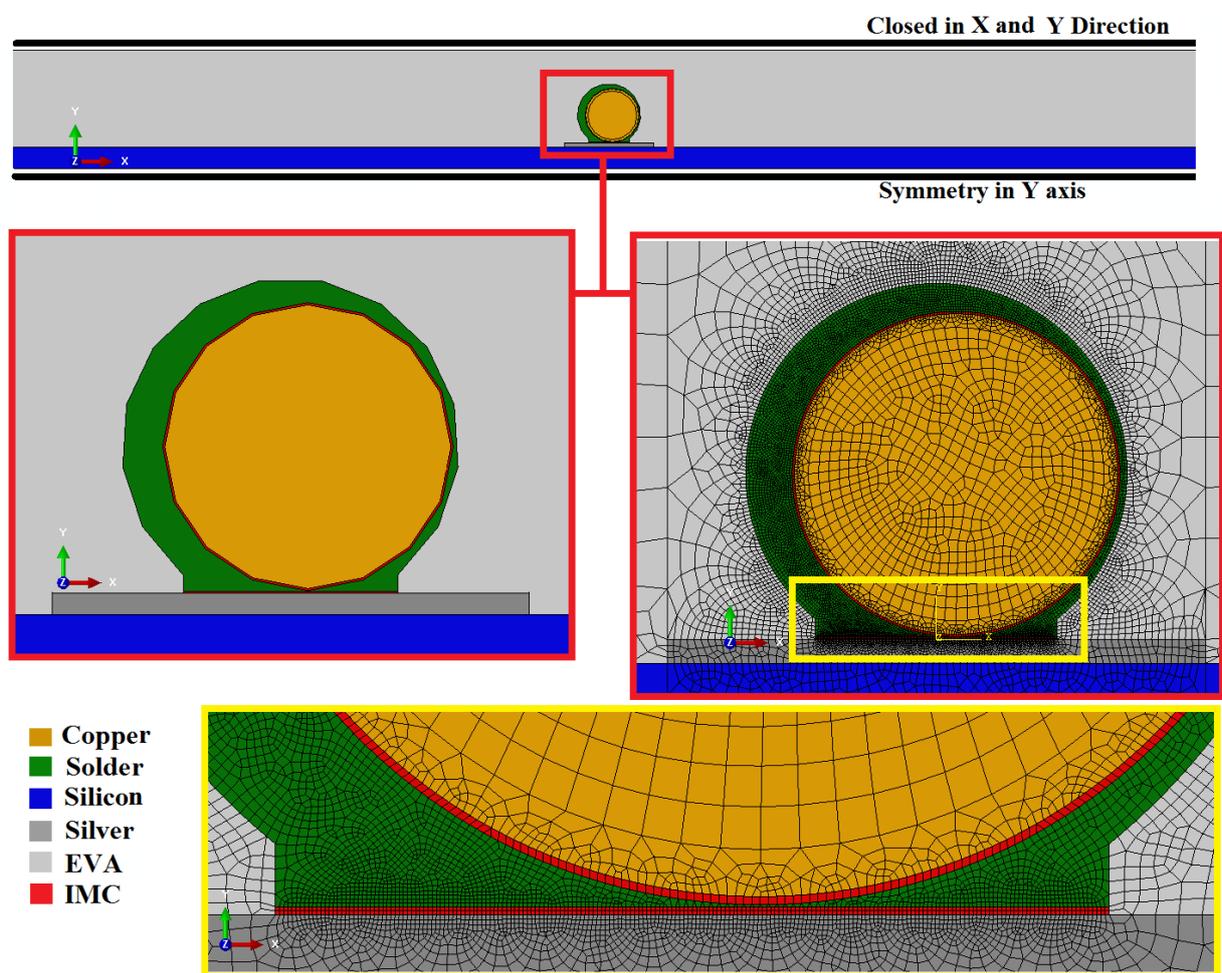
104 In this study, 2D symmetrical simulation of interconnection strips with four-node plane strain  
105 elements (CPE4) is performed as this requires less Computational processing and storage compared to  
106 3D simulation modelling. Altogether, 82 interconnection strip models were simulated in ABAQUS  
107 CAE.

108 Based on Multi-Busbar Connector Prototype specifications published by (Walter et al., 2014), the  
109 diameter of copper ribbon and width of silver pad are assumed to be 270  $\mu\text{m}$  and 450  $\mu\text{m}$ ,  
110 respectively; and the thickness of silver pad and silicon cell are 20  $\mu\text{m}$  and 220  $\mu\text{m}$  (Note: in order to  
111 achieve stable response, the gaps between the interconnection strips were considered to be 6mm). To  
112 investigate effect of geometrical parameters, four different solder coating thickness (15  $\mu\text{m}$ , 20  $\mu\text{m}$ ,  
113 25  $\mu\text{m}$  and 30  $\mu\text{m}$ ) were evaluated. In addition, the effect of IMC layer thickness (silver pad interface)  
114 was investigated; with four different IMC layer thickness considered (1  $\mu\text{m}$ , 2  $\mu\text{m}$ , 3  $\mu\text{m}$  and 4  $\mu\text{m}$   
115 was used to reflect wide range of soldering temperature). However, the IMC layer thickness on the  
116 copper wire core interface is assumed to be 2  $\mu\text{m}$  and symmetry boundary conditions are applied for  
117 the bottom of silicon cell. The top edge of Ethylene Vinyl Acetate (EVA) which in contact with the  
118 glass protective sheet in PV Module is considered closed in X and Y direction (i.e. no X and Y  
119 direction movement is assumed). The temperature of whole solar cell has been increased linearly and  
120 isothermally to the crack initiation temperature. Figure 2, shows the discretization and meshing  
121 method used in the study and the configuration for the Copper core, Solder Joint, Intermetallic  
122 Compound (IMC), Silver Pad, EVA and Silicon cell. A very fine mesh size of 1  $\mu\text{m}$  was used for  
123 investigating the effect of the IMC layer thickness on the silver pad interface; and to determine the  
124 crack initiation temperature and location of micro-crack.

125 To meet the requirements of the theory of Linear Elastic Fracture Mechanics (LEFM) which is  
126 basis of XFEM for investigation of crack propagation, the IMC layer between solder and silver pad is  
127 assumed as a brittle elastic material; whilst all other metallic materials in the PV Module (silver,  
128 copper and solder) are considered to have plastic behaviour. Based on the models for traction-

129 separation laws, different of types of strain and stress components (MAXPS / MAXPE<sup>2</sup>, MAXS /  
 130 MAXE<sup>3</sup> and QUADS/ QUADE<sup>4</sup>) can be used to control damage initiation. In this study, the  
 131 maximum nominal stress (MAXS) of Shear and Tensile components is considered as the controlling  
 132 parameters for the damage initiation. This means that the damage (crack) is initiated when these  
 133 components exceed the defined limits. Then, the initiated crack will be evaluated as Fracture Energy  
 134 formulations which can be formed by the parameter of Fracture Toughness and Elastic Modulus (Du,  
 135 2009).

136 Table 1 shows the mechanical properties of materials used in the simulation. The temperature  
 137 dependency of the thermal coefficient of expansion was considered for the Silver, Solder, IMC and  
 138 copper materials; and the temperature dependency of the Young's Modulus and plastic behaviour was  
 139 also considered for Solder as shown in Table 2.



140 Figure 2 Applied mesh and material arrangement of the PV module Cell in the top interconnecting area  
 141

142

<sup>2</sup> Maximum principal stress (MAXPS) and maximum principal strain (MAXPE)

<sup>3</sup> Maximum nominal stress (MAXS) and maximum nominal strain (MAXE)

<sup>4</sup> Quadratic nominal stress (QUADS) and quadratic nominal strain (QUADE)

143

Table 1 Mechanical Properties of Material used in the FEM simulation of strip interconnection

Parameter (unit)	Material					
	IMC (SnAg)	Solder Joint	Silver (AZoM, 2001)	Copper (Jing, et al., 2015)	EVA (Department, 2003)	Silicon (Owen- Bellini et al., 2015)
Elastic Modulus (MPa)	See Table 2	See Table 2	69	121	11	170
Poisson's Ratio (-)	0.35	0.35	0.365	0.34	0.499	0.28
Yield Stress (MPa)	-	-	43	121	12	170
Shear Strength (MPa) (Deng et al., 2005)	27.6- 1.95*H <sub>IMC</sub>	30	-	-	-	-
Tensile Strength (MPa) (Zhong et al., 2010)	65	78	-	-	-	-
Fracture Toughness (MPa.m <sup>0.5</sup> )	7	7	40	30	-	0.83
Thermal Expansion Coefficient (ppm/k)	See Table 2	See Table 2	See Table 2	See Table 2	270	2.6
Plastic Stress- Strain Curve (MPa)	-	See Table 2	43@0.001 120@0.04	121@0.001 186@0.004 217@0.01 234@0.02 248@0.04	-	-

144

Table 2 Temperature dependent properties of interconnection materials

Temp. (c)	CTE for Copper (ppm/k) (Taulaukian et al., 1975) (Interpolated)	CTE for Silver (ppm/k) (Taulaukian et al., 1975) (Interpolated)	CTE for Solder (ppm/k) (49-0.07T) (Li, el al., 2012)	Solder Young's Modulus (MPa) (21.3+0.017T) ( Li et al., 2012)	Stress (MPa) (at 0.0, 0.065 Plastic Strain) (Siviour et al., 2005) (Interpolated)
0	16.22	18.67	21.3	49	71, 145
30	16.60	18.98	21.81	46.9	52, 131
60	16.91	19.20	22.32	44.8	16, 110
90	17.22	19.42	22.83	42.7	-
120	17.53	19.65	23.34	40.6	-
150	17.76	19.91	23.85	38.5	-

145

146

147

148

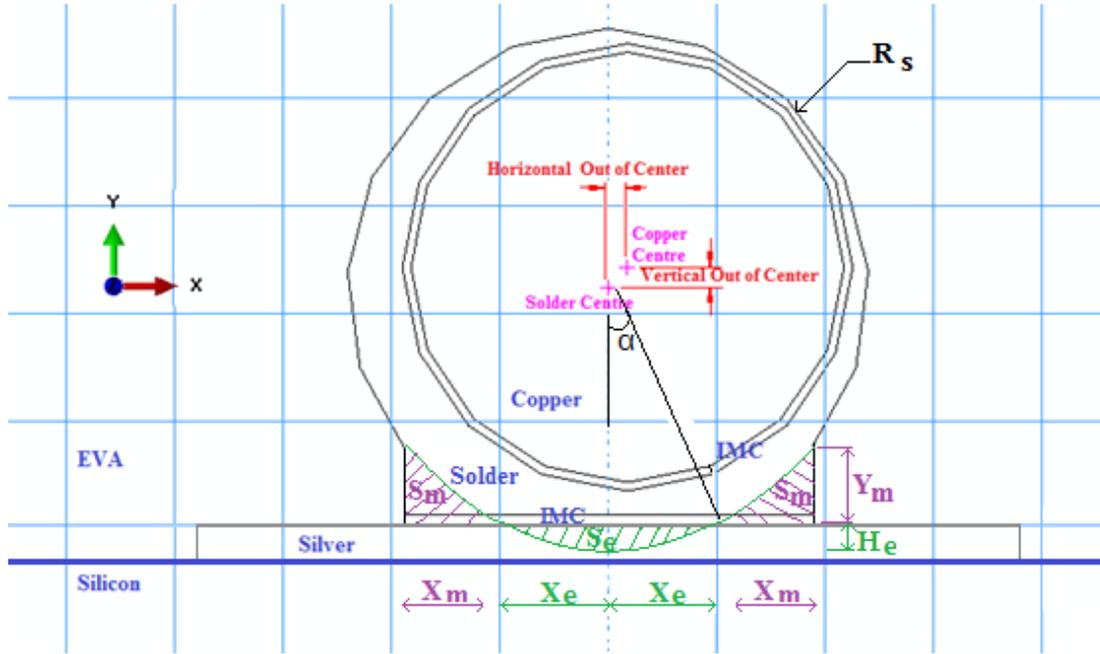
149

150

151

The schematic view of cross section of the round ribbon interconnector presented in Figure 3, shows that  $S_e$  (the molten solder area) located between the lower end of strip interconnection and the silver pad is displaced sideways to the both left and right corners of interconnect strip; thereby increasing the contact area between the strip interconnection and silver pad. The mathematical expressions used for calculating the extra contact length  $X_m$  between the solder joint and the Silver pad is given by equation eq. #5. If we assume that the Solder Area  $S_e$  with height  $H_e$  is sub-divided into triangular areas  $S_m$ ; then the derivation of the extra contact length  $X_m$  is as detailed in eq. #1 - #5.

152 To investigate the effect of solder joint height on the strength of the strip interconnection, four  
 153 different molten solder heights  $H_e$  was considered (i.e.  $H_e$  was varied from 25%, 37.5%, 50% to  
 154 62.5% of solder coating thickness).



155  
 156 Figure 3 Cross section of round ribbon interconnector with view of vertical and horizontal out of centre

$$Eq. 1: X_e = \sqrt{R_s^2 - (R_s - H_e)^2}$$

$$Eq. 2: \tan(\alpha) = \frac{X_e}{R_s - H_e} = \frac{Y_m}{X_m}$$

$$Eq. 3: S_e = R_s^2 \cdot \alpha - 2 \cdot X_e \cdot (R_s - H_e)$$

$$Eq. 4: S_m = \frac{Y_m \cdot X_m}{2} = \frac{S_e}{2}$$

$$Eq. 5: X_m = \sqrt{\frac{S_e}{\tan(\alpha)}} = \frac{R_s^2 \cdot \text{Arctan}\left(\frac{\sqrt{R_s^2 - (R_s - H_e)^2}}{R_s - H_e}\right) - \sqrt{R_s^2 - (R_s - H_e)^2} \cdot (R_s - H_e)}{\sqrt{\frac{R_s^2 - (R_s - H_e)^2}{R_s - H_e}}}$$

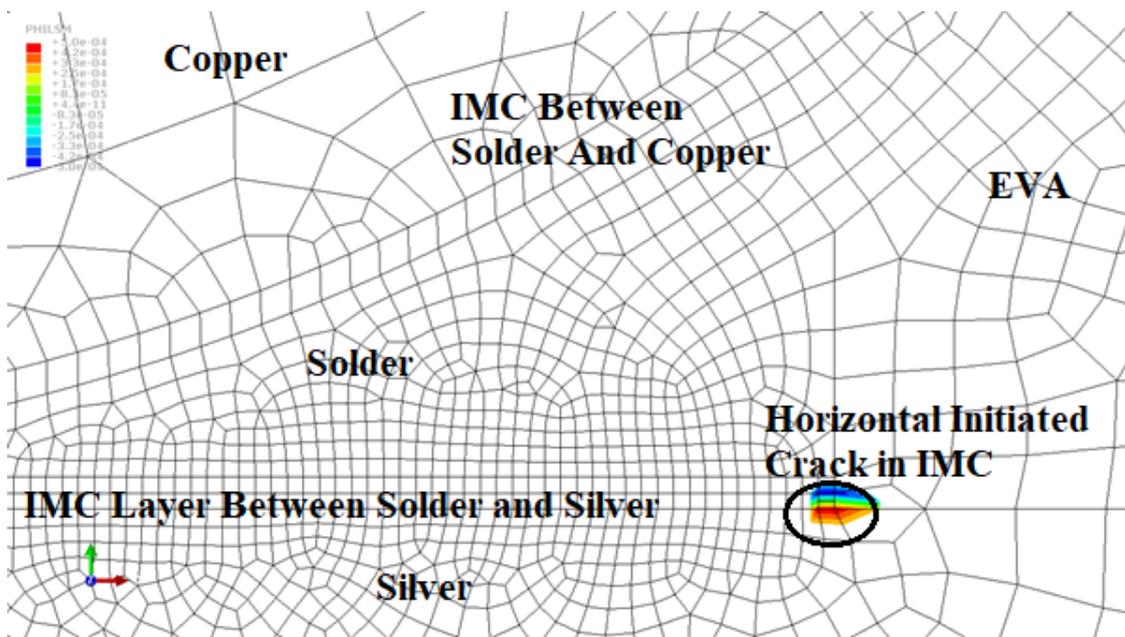
### 157 3. Results and Discussions

158 This section presents the results from the study on the investigation of the effect of solder coating  
 159 non-homogeneity on the thermomechanical response of round copper wires used for PV module  
 160 interconnections and the evaluation of the effect of three geometrical parameters, namely the  
 161 Intermetallic Compound (IMC) layer thickness, solder joint height and thickness on the strength of the  
 162 interconnections.

163 3.1. Crack location

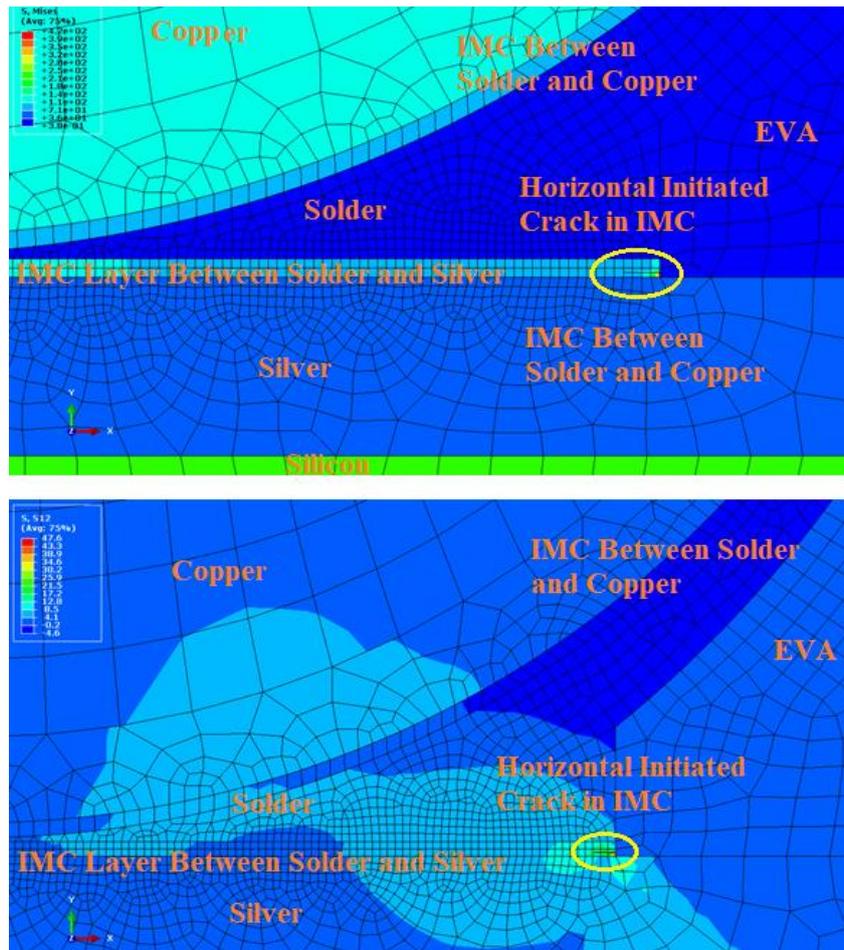
164 The results of the XFEM analysis presented in Figure 4 shows the one-micrometre crack that is  
165 initiated in the IMC layer between the solder joint material and silver pads. For all the geometrical  
166 parameters considered in the study, the results show that the highest stress accumulation (and the  
167 micro-crack initiation), occurs at the edge of the IMC layer interface between the solder joint and  
168 silver pad. . The results also show that the crack propagation is in the parallel direction to the silver  
169 pad. The material properties presented in Table 2, show that the micro-crack at the interface between  
170 the solder joint and silver pad is caused by the high coefficient of thermal expansion (CTE) mismatch  
171 between solder joint and silicon layer.

172 Figure 5 shows the Von Mises stress and shear stress contours in the IMC layer between solder  
173 coat and Silver pad at the micro-crack initiation temperature and the crack propagation. Although the  
174 concentration of shear stress in the IMC layer leads to the micro-crack initiation and propagation  
175 along the IMC interface layer; further crack progression can be caused by thermal recycling in hot  
176 temperature operation which can lead to the disconnection of the strip interconnection from the Silver  
177 pad.



178  
179

Figure 4 crack initiation in IMC layer between solder coat and Silver pads

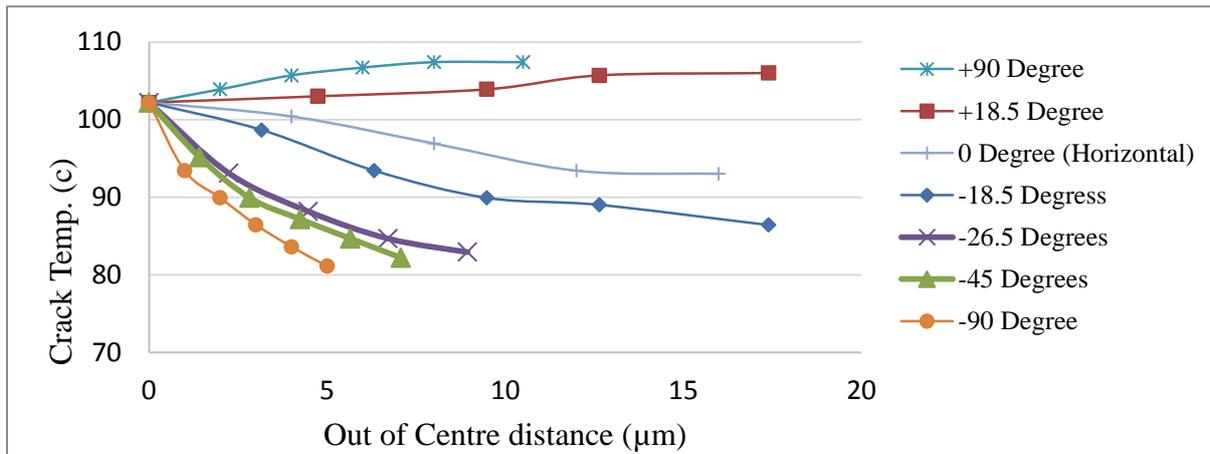


180  
 181 Figure 5 Von Mises Stress (Top) and Shear Stress (Below) Contours in temperatures of micro-cracking of IMC  
 182 layer between solder coat and Silver pads

### 183 3.2. Out of centre positioning of copper in the interconnection strip

184 Figure 6 shows the crack initiation temperatures for different geometries and out of centre  
 185 distances in different directions including the upward Y direction (+90 Degrees), the downward Y  
 186 direction (-90 Degrees), the horizontal (0 Degree) and also the upward and the downward radial  
 187 directions (-9.5 and +/- 18.5 Degrees). The molten solder height in this graph is considered 50% of  
 188 solder thickness (10  $\mu\text{m}$ ) and the thickness of IMC layer is assumed to be 2  $\mu\text{m}$ . The results of the  
 189 simulations show that by increasing the out of centre distance (i.e. with more non-homogenous solder  
 190 coating) in downward directions, the crack initiation strength of the solder joint decreases (Note: the  
 191 highest rate is for -90 Degree out of centre positioning). This means that for the downward out of  
 192 centre positioning of the copper inside the coated wire, the crack initiation temperature decreases with  
 193 increasing non-homogeneity of coating. This is because the narrow solder thickness between copper  
 194 wire and silicon layer results in an increase in the accumulated strain and thermal stress due to the  
 195 high thermal coefficient of expansion (CTE) mismatch between the solder and silver materials.. For  
 196 this reason, the micro-cracks are more likely to occur at the interface between the IMC layer and the  
 197 silver pad. The results also show that for the upward out of centre positioning of the copper inside the  
 198 coated wire, the cracking initiation strength of the solder joint is increases with more non-

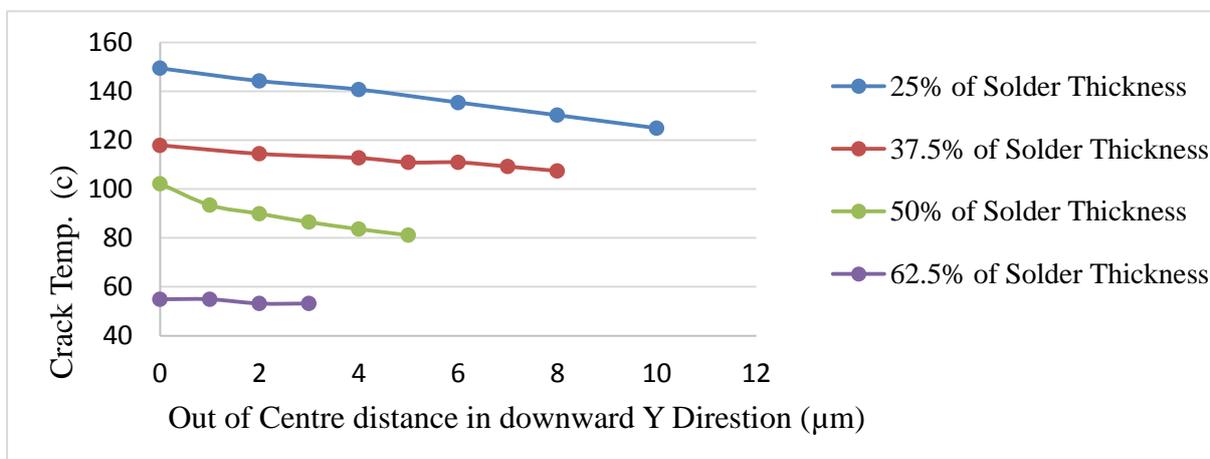
199 homogenous solder coating (increases the micro-crack initiation strength); and the crack initiation  
 200 temperature is also much higher than those for downward out of centre positioning directions.



201 Figure 6 Temperature of crack initiation for different direction of out of centre distances. The molten solder  
 202 height is 50% of Solder thickness and the thickness of IMC layer is assumed to be 2 µm.  
 203

### 204 3.3. Molten Solder Height

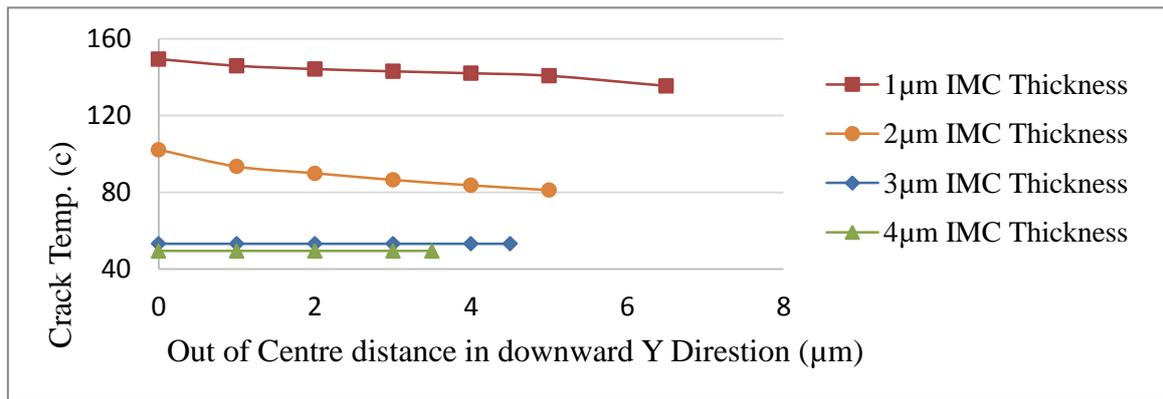
205 Figure 7 shows the effect of the out of centre distance in the downward Y direction (-90 Degrees)  
 206 for different molten solder heights on the crack initiation temperature. In principle, increasing the  
 207 molten solder height increases the solder contact between the copper wire and the silver pad and  
 208 hence lead to more reliable connection; the results in Figure 7 suggests that by increasing the molten  
 209 solder height and increasing solder contact will actually result in micro-cracks occurring at much  
 210 lower temperatures. This reduction in the micro-crack initiation temperature can be attributed to the  
 211 high levels of thermal stress resulting from the coefficient of thermal expansion (CTE) mismatch in  
 212 the increased solder contact. However, the reduction in the micro-crack initiation temperature due out  
 213 of centring is mitigated by the increase in solder joint strength; and thereby reducing the impact of out  
 214 of centring with high molten solder height.



215 Figure 7 Temperature of crack initiation via out of centre distances in downward Y direction for different  
 216 molten solder height. Thickness of IMC layer is considered 2 µm.  
 217

218 3.4. Thickness of IMC Interface Layer between strip interconnection and silver pad

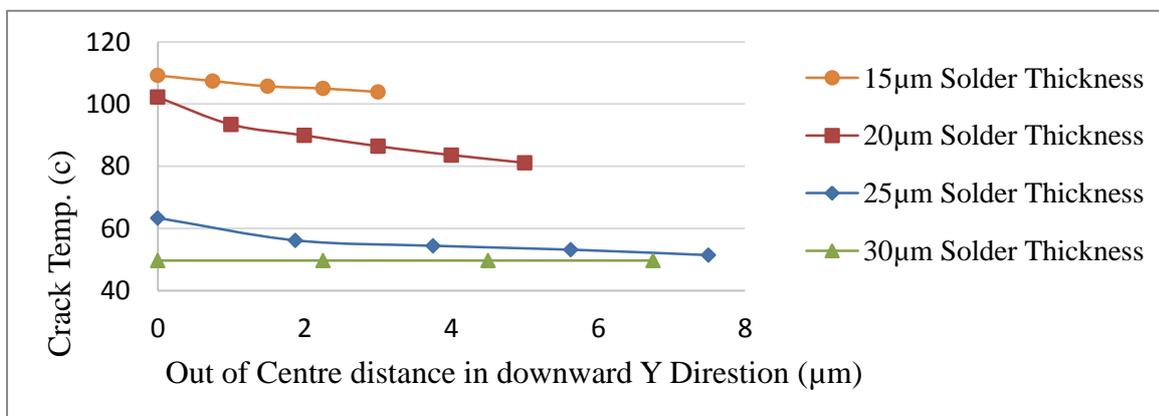
219 Figure 8 demonstrates the effect of out of centre distances in downward Y direction (-90 Degrees)  
220 on crack initiation temperature for cases where the solder thickness is 20  $\mu\text{m}$  and the molten solder  
221 height is 10  $\mu\text{m}$  (i.e. 50% of solder thickness), with different IMC layer thickness between strip  
222 interconnection and silver pad. The results show that solder joints with lower IMC layer thickness  
223 exhibited higher strengths than solder joints with higher IMC thickness. The results also show that  
224 there is a reduction in the crack initiation temperature with increases in the out of centre positioning;  
225 however this is negligible with higher IMC layer thickness.



226 Figure 8 Temperature of crack initiation via out of centre distances in downward Y direction for different IMC  
227 layer thickness. The molten solder height is 50% of Solder thickness.  
228

229 3.5. Solder Coating Thickness

230 Figure 9 shows the effect of out of centre distances on crack initiation temperature in downward  
231 Y direction (-90 Degrees) for cases with 10  $\mu\text{m}$  molten solder height (50% of Solder thickness), IMC  
232 layer thickness of 2  $\mu\text{m}$  with different solder thickness. The result shows that for higher solder  
233 thickness, increasing the solder thickness reduces the crack initiation temperature; but the effect of out  
234 of centre distance on crack initiation temperature is negligible.



235 Figure 9 Temperature of crack initiation via out of centre distances in downward Y direction for different solder  
236 thickness. The molten solder height is 10  $\mu\text{m}$  and the IMC layer thickness is 2  $\mu\text{m}$ .  
237

#### 238 **4. Conclusion**

239 This study presents the results of the Extended Finite Element Method (XFEM) Simulation  
240 performed on the round wire used in Multi-Busbar (MBB) strip interconnection of solar PV module to  
241 evaluate micro-crack initiation in non-homogenous solder coating at high temperatures. The  
242 determination of the conditions for micro-crack initiation in the solder coating on the round wire is  
243 important for determining the interconnection design parameters that will ensure long term reliability  
244 of the solar PV module. The results from the simulation study would be used for MBB  
245 interconnection design and for the evaluation of solar PV module performance and reliability. The  
246 study evaluates the two main parameters of non-homogeneity (out of centre value and direction), and  
247 also investigates the effect of geometrical parameters including Intermetallic Compound (IMC) layer  
248 thickness and solder joint height. The XFEM in ABAQUS 6.17 software package was used to  
249 determine the hot temperature required for micro-crack initiation for each joint design. Also, XFEM  
250 analysis was used to predict the location of the micro- crack in the IMC layer between solder coating  
251 and silver pad.

252 The results show that the micro-crack is initiated in the parallel direction with silver pad and the  
253 initiation temperature is most affected by the direction of solder coating non-homogeneity and the  
254 downward vertical direction of out of centre positioning of copper in the solder coating leads to the  
255 most reduction in crack initiation temperature (up to 21% reduction was observed for the case 5.5  $\mu\text{m}$   
256 out of centre distance). The results also show that by increasing the amount of molten solder/solder  
257 thickness, the micro-crack initiation will take place at a lower temperature due to high thermal stress  
258 accumulation in the IMC layer. Also, it was found that at high solder thickness the micro-crack  
259 initiation temperature is less affected by non-homogeneity. The results also show that increasing the  
260 IMC layer thickness leads to a decrease in the micro-crack initiation temperature threshold and  
261 thinner IMC layers are more sensitive with non-homogeneity. The results of the study will also be  
262 useful for researchers in evaluating the impact of IMCs layers in the solder region boundaries on solar  
263 PV module reliability and in developing design-for-reliability guidelines.

#### 264 **References**

265 ABAQUS, I., 2017. *ABAQUS User's and Theory Manuals; Version 6.17*, s.l.: ABAQUS, Inc.: Providence  
266 Rhode Island, RI, USA.

267 AZoM, 2001. *Silver - Applications and Properties of Silver*. [Online]  
268 Available at: <https://www.azom.com/properties.aspx?ArticleID=600>

269 Braun, S. et al., 2013. *The multi-busbar design: an overview*. s.l., s.n., p. 86 – 92.

270 Braun, S., Micard, G. & Hahn, G., 2012. *Solar cell improvement by using a multi busbar design as*  
271 *front electrode*. s.l., s.n., pp. 227-233.

272 Deng, X., Sidhu, R. j. & Chawla, N., 2005. Influence of reflow and thermal aging on the shear strength  
273 and fracture behavior of Sn-3.5Ag Solder/Cu joints. *Metallurgical and Materials Transactions A* 36(1),  
274 pp. 55-64.

275 Department, C. U. E., 2003. *Materials Data Book*. Cambridge, UK: Cambridge University Engineering  
276 Department Data Books.

277 Du, Z.-z., 2009. *eXtended Finite Element Method (XFEM) in Abaqus*. [Online]  
278 Available at: <http://www.simulia.com/download/rum11/UK/Advanced-XFEM-Analysis.pdf>

279 Itoh, U. et al., 2014. *Solder joint failure modes in the conventional crystalline Si module*. s.l., s.n.

280 Jeong, J.-S., Park, N. & Han, C., 2012. Field failure mechanism study of solder interconnection for  
281 crystalline silicon photovoltaic module. *Microelectronics Reliability*.

282 Jing, X. et al., 2015. *Effect of pre-CMP annealing on TSV pumping in thermal budget and reliability*  
283 *test*. Hsinchu, Taiwan, IEEE 22nd International Symposium on the Physical and Failure Analysis of  
284 Integrated Circuits.

285 Karppinen, J., Laurila, T., Ka, J. & Li, J., 2012. Reliability of Lead-Free Solder Interconnections in  
286 Thermal and Power Cycling Tests. *IEEE TRANSACTIONS ON COMPONENTS AND PACKAGING*  
287 *TECHNOLOGIES, VOL. 32, NO. 2*.

288 Mohammadi, S., 2008. *Extended Finite Element Method: For Fracture Analysis of Structures*.  
289 s.l.:Wiley/Blackwell.

290 O. Ogbomo, O., H. Amalu, E., Ekere, N. & P.O., O., 2018. Effect of operating temperature on  
291 degradation of solder joints in crystalline silicon photovoltaic modules for improved reliability in hot  
292 climates. *Solar Energy* 170, p. 682–693.

293 Owen-Bellini, M., Zhu, J., R. Betts, T. & Gottschalg, R., 2015. *Thermo-Mechanical Stresses of Silicon*  
294 *Photovoltaic Modules*. Bangor, United Kingdom, s.n.

295 Pareek, S., Chaturvedi, . N. & Dahiya, R., 2017. Optimal interconnections to address partial shading  
296 losses in solar photovoltaic arrays. *Solar Energy* 155, p. 537–551.

297 Pecht, M., 1993. *Soldering processes and equipment*. s.l.:Wiley-IEEE. p. 18. ISBN 978-0-471-59167-2. .

298 Rendler, L. C. et al., 2016. *Mechanical stress in solar cells with multi busbar interconnection-*  
299 *Parameter study by FEM simulation*. s.l., s.n.

300 Rendler, L. C., Walter, J., Goldenberg, S. & Bein, A., 2018. Mechanical and electrical properties of  
301 wave-shaped wires for low-stress interconnection of solar cells. *Solar Energy Materials and Solar*  
302 *Cells* 176, p. 204–211.

303 Schindler, S. et al., 2013. *Soldering process and material characterization of miniaturized contact*  
304 *structures of a newly developed multi busbar cell metallization concept*. s.l., s.n., p. 480–483.

305 Schneider A, Rubin L & Rubin G, 2006. *Solar Cell Efficiency Improvement by New Metallization*  
306 *Techniques - the Day4 Electrode Concept*. Waikeolu, Hawaii, USA, 4th IEEE World Conference on  
307 Photovoltaic Energy Conversion.

308 Sivakumar, G. & Maji, V., 2016. Simulation of crack propagation in rocks by XFEM. *Recent Advances*  
309 *in Rock Engineering (RARE 2016)*.

310 Siviour, C. R., Walley, S., Proud, W. & Field, J., 2005. Mechanical properties of SnPb and lead-free  
311 solders at high rates of strain. *Journal of Physics D: Applied Physics*, p. 4131–4139.

312 Song, H.-J. et al., 2019. Conductive paste assisted interconnection for environmentally benign  
313 leadfree ribbons in c-Si PV modules. *Solar Energy* 184, pp. 273-280.

314 Tae-hee Jung, H.-e. S. H.-k. A. G.-h. K., 2014. A mathematical model for cell-to-module conversion  
315 considering mismatching solar cells and the resistance of the interconnection ribbon. *Solar Energy*  
316 103, p. 253–262.

317 Taulaukian, Y., Kirby, R., Taylor, R. & Desai, P., 1975. *Thermal expansion Metallic Elements and*  
318 *Alloys-THERMOPHYSICAL PROPERTIES OF MATTER. VOLUME 12*. New York: SPRINGER  
319 SCIENCE+BUSINESS MEDIA, LLC.

320 Walter, J. et al., 2014. *Multi-wire interconnection of busbar-free solar cells*. s.l., s.n., p. 380–388.

321 Zarmai, M. T., Ekere, N., Oduoza, C. & H. Amalu, E., 2016. Optimization of thermo-mechanical  
322 reliability of solder joints in crystalline silicon solar cell assembly. *Microelectronics Reliability*, p. 117–  
323 125.

324 Zhong, W., QIN, F., AN, T. & WANG, T., 2010. *Mechanical Properties of Intermetallic Compounds in*  
325 *Solder Joints*. s.l., s.n.

326